

AWAD 2013



**2013 Asia-Pacific Workshop on Fundamentals and
Applications of Advanced Semiconductor Devices**

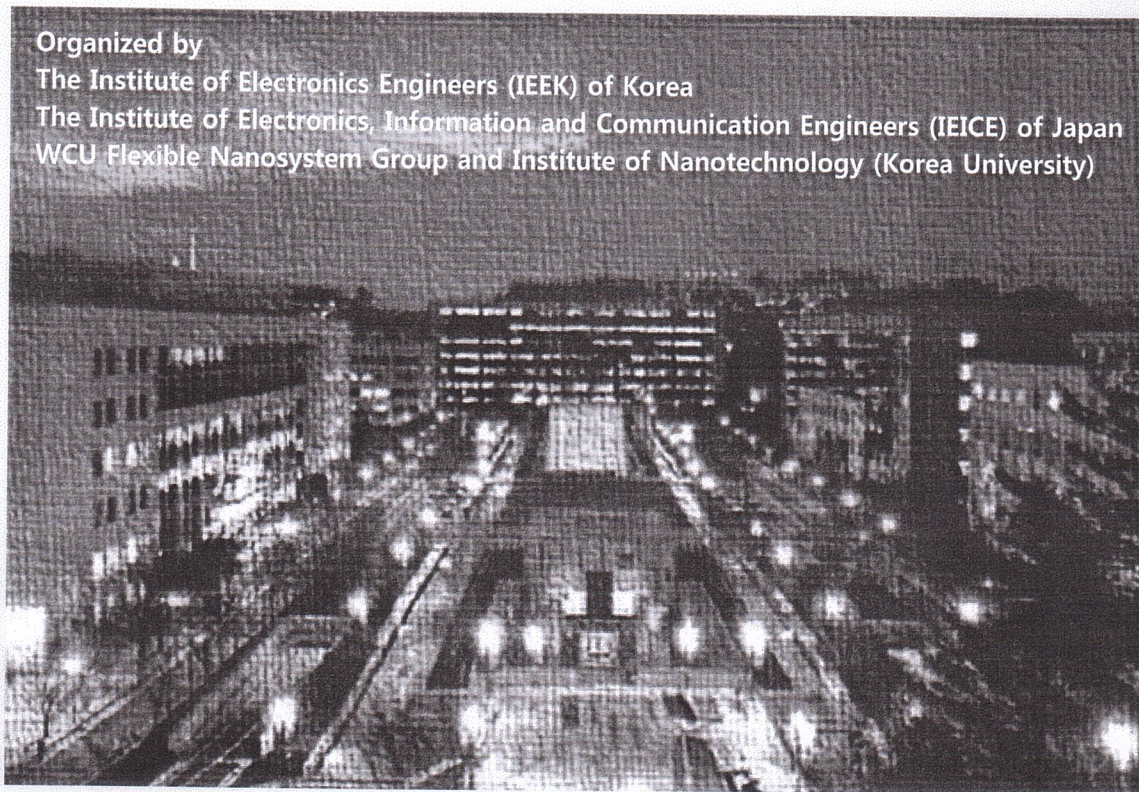
**June 26-28, 2013
Korea University, Seoul, Korea**

Organized by

The Institute of Electronics Engineers (IEEK) of Korea

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Session 2B: Reliability and Characterization I

Session Co-Chairs: Woo Young Choi (Sogang Univ.), Rihito Kuroda (Tohoku Univ.)

2B-1 13:30 – 14:00

Invited talk : Density-of-States Based Modeling of Oxide Thin-Film Transistors: Toward Instability-Aware Design

Dae hwan Kim (Kookmin University)

2B-2 14:00 – 14:15

Proton irradiation effects on p- and n-type GaN

H. Okada, Y. Okada, H. Sekiguchi, A. Wakahara (Toyohashi University), S. Sato and T. Ohshima (Japan Atomic Energy Agency)

2B-3 14:15 – 14:30

Impact of Injected Carrier Types to Stress Induced Leakage Current Using Substrate Hot Carrier Injection Stress

H. W. Park, A. Teramoto, T. Inatsuka, R. Kuroda, S. Sugawa and T. Ohmi (Tohoku University)

Poster Session 14:45 – 15:30

Session Chair: Woo Young Choi (Sogang Univ.)

P-1 GaN epitaxy layer grown on Si (111) substrate using Ge buffer layer for extended CMOS integration

Yeon-Ho Kil, Young-Jae Park, Taek Sung Kim, Chang-Hee Hong, Chel-Jong Choi, and Kyu-Hwan Shim (Chonbuk National University)

P-2 Delay time component of InGaAs MOSFET caused by dynamic source resistance

Masayuki Yamada, Yasuyuki Miyamoto (Tokyo Tech.), and Ken Uchida (Keio Univ.)

P-3 Optimization of Mo-based, Au-free Ohmic contact for AlGaIn/GaN-on-Si Heterostructure Field Effect Transistors

Shinhyuk Choi, Jae-Gil Lee, Hyungtak Kim and Ho-Young Cha (Hongik Univeristy)

P-4 Electrical Properties of Au/n-Ge Schottky Barrier Diode With Graphene Interlayer

Zagarzusem Khurelbaatar, Chel-Jong Choi, Gil-Song Kim, Daoheung Bouangeune, Keun-Soo Kim, Jung Tae Nam, Sang-Kwon Lee, Kyu-Hwan Shim (Chonbuk National University)

P-5 Electrostatic Discharge Natures for Transient Voltage Suppression Diodes Made of n+p-p+ and p+n-n+ Junctions

Daoheung Bouangeune, Zagarzusem Khurelbaatar, Sang-Sig Choi, Chel-Jong Choi, Deok-Ho Cho, and Kyu-Hwan Shim (Chonbuk National University)

P-6 Bidirectional Transient Voltage Suppression Diodes for the Protection of High Speed Data

Density-of-States Based Modeling of Oxide Thin-Film Transistors: Toward Instability-Aware Design

Dae Hwan Kim,^{†‡} Kyung Min Lee[†], Sung-Jin Choi[†], and Dong Myong Kim[†]

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Abstract The negative bias illumination stress (NBIS) has emerged as a serious hurdle to the successful industrialization and mass production of amorphous oxide thin-film transistor (TFT) technology. In this work, the subgap density-of-states (DOS) of amorphous In-Ga-Zn-O (a-IGZO) TFTs was extracted by combining the optical C - V and the generation-recombination current spectroscopy with the I - V model, named as DeAOTS (Density-of-states based Amorphous Oxide semiconductor Thin-film transistor Simulator). We also incorporated the recently proposed peroxide formation theory into our DOS model and simulation platform. Finally it was found that our DOS-based instability model reproduced the measured NBIS time-evolution of I - V characteristics of a-IGZO TFTs very well. Our results have the potential of playing a powerful role in the instability-aware design of oxide TFTs without a long-term stress and/or reliability test.

Keyword oxide thin-film transistor, density-of-states, instability model, peroxide, negative bias illumination stress.

1. Introduction

Amorphous oxide thin-film transistor (TFT) has been recognized as one of promising candidates substituting the amorphous silicon (a-Si), low temperature polycrystalline silicon (LTPS) and organic TFTs as switching/driving devices in the active-matrix liquid crystal displays (AMLCDs) and/or the active-matrix organic light emitting diode displays (AMOLEDs), because of its advantages of the flexibility, visible light transparency, large-area uniformity of a low temperature sputter-deposited amorphous material, and quite high carrier mobility (Table 1). In real, very recently, various display backplanes driven by oxide TFTs have been competitively demonstrated [1]-[3]. Especially, the compatibility of bottom gate-structured oxide TFTs with the a-Si TFT-based TFT-LCD manufacturing line makes them promising solutions as the cost-effective driving devices for AMOLEDs in comparison with top gate-structured LTPS TFTs. However, the photoelectric stress-induced instabilities have emerged as challenging issues for industrialization of the oxide TFT-driven display backplanes [4], [5]. Therefore, the fundamental understanding and quantitative projection of instability became more important and indispensable for the systematic design of the oxide TFT materials, devices, and circuits. In specific, the negative bias illumination stress (NBIS)-induced instability needs to be intensively analyzed because in either AMLCDs or AMOLED display backplane, switching TFTs are commonly subject to the

negative gate-to-source voltage (V_{GS}) and positive drain-to-source voltage (V_{DS}) stress under photo-illumination during most of each frame time [4].

Table 1 Comparison of TFT active materials

TFT active material	a-Si	LTPS	Oxide
Field effect mobility	$\sim 1 \text{ cm}^2/\text{Vs}$	$\sim 100 \text{ cm}^2/\text{Vs}$	$> 10 \text{ cm}^2/\text{Vs}$
TFT uniformity	good	poor	good
Transparency	poor	poor	good
Process temperature	$\sim 250^\circ\text{C}$	$> 250^\circ\text{C}$	$> \text{RT}$
Fabrication technique	PECVD	PECVD	sputtering
Channel type	NMOS	PMOS (CMOS)	NMOS
Cost	low	high	low
Yield	-	medium	medium

In this work, we quantitatively investigated the NBIS-induced threshold voltage shift (ΔV_T) in amorphous In-Ga-Zn-O (a-IGZO) TFTs and established its physical model and simulation platform based on experimentally extracted density-of-states (DOS). In order to extract subgap DOS of TFT active material and calculate the TFT I - V characteristics, the optical C - V spectroscopy [6], the generation-recombination current (I_{G-R}) spectroscopy [7], and the improved DeAOTS (Density-of-states based Amorphous Oxide semiconductor Thin-film transistor Simulator) [8], [9] were conjointly used. Finally, it was

found that our peroxide formation [10]-based instability model successfully reproduced the NBIS time (t_{NBIS})-evolution of the measured I - V characteristics.

2. DOS Extraction

The subgap DOS in oxide semiconductor thin-film which acts as the active material of oxide TFTs can be schematically illustrated as shown in Fig. 1. Among DOS parameters, the $g_A(E)$ [$\text{cm}^{-3}\text{eV}^{-1}$], *i.e.*, the acceptor-like states near the conduction band minimum (CBM: E_C), which are known to originate from the disorder of metal cation s -bands, namely, In-O-metal bonding angle variation in the case of a-IGZO and denote the degree of strong dispersion of electronic structures near E_C [11]-[13], determine the I - V characteristics including the subthreshold swing (SS). On the other hand, the $g_{\text{TD}}(E)$ [$\text{cm}^{-3}\text{eV}^{-1}$], *i.e.*, the donor-like states near the valence band maximum (VBM: E_V), originate either from the amorphous randomness, namely, the oxygen p -band disorder (intrinsic) or from the deep level donors (extrinsic). While the former reflects the degree of strongly localized states near E_V , namely, a weak dispersion of holes, the latter can be identified by oxygen vacancy (V_O) defects with the inward relaxation with neighboring metal atoms followed by the large vacancy size [14]. The $g_{\text{TD}}(E)$ is closely related to the photoelectric stress-induced instability [4] and the N_{SD} [cm^{-3}] works as the density of shallow donors, *i.e.*, n -type doping concentration, for supplying free carriers. In addition, the V_O s of oxide films indirectly but strongly affects the concentration of shallow donors (N_{SD}) by transforming them to shallow donors through their interaction either with metal cations [15] or with hydrogen species [16], although they are energetically favored to be deep donors [17], [18].

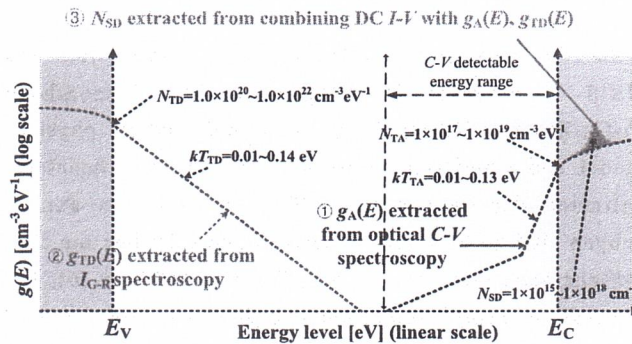


Fig. 1 The schematic illustration of subgap DOS with its extraction technique.

The used a-IGZO TFTs had the etch stopper-type and were integrated with circuits integrated on a glass substrate as shown in Fig. 2. The bottom-gate and top contact structured TFTs were fabricated using the standard semiconductor processes. The detailed process was given in [9].

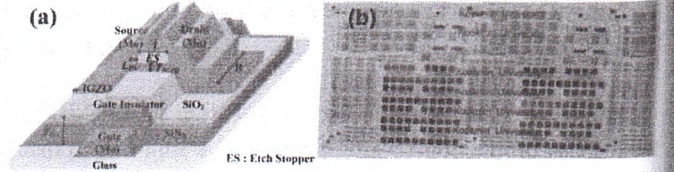


Fig. 2 (a) A schematic illustration of integrated a-IGZO TFT. (b) The photograph of a-IGZO TFTs and circuits integrated on a glass substrate [9].

The fabricated a-IGZO TFT has a gate dielectric thickness $T_{\text{OX}}=258$ nm as an equivalent oxide thickness (EOT) (400/50 nm of $\text{SiN}_x/\text{SiO}_x$ bilayer), the channel width $W=200$ μm , the channel length $L=100$ μm , the gate-to-S/D overlap length $L_{\text{ov}}=10$ μm , and the thickness of a-IGZO thin-film $T_{\text{IGZO}}=50$ nm. Typical device parameters were obtained to be the ON current $I_{\text{DS}}=1.13$ [μA] at $V_{\text{GS}}=V_{\text{DS}}=10.1$ [V], $V_{\text{ON}}=2.8$ [V] (V_{ON} was defined as V_{GS} at $I_{\text{DS}}=1$ [pA] in the transfer curve), the threshold voltage $V_{\text{T}}=7.2$ [V] (defined as V_{GS} at $I_{\text{DS}}=10$ [nA] in the transfer curve), the SS=0.74 [V/dec], the field-effect mobility (μ_{FE})=9.4 [cm^2/Vs] at $V_{\text{GS}}=20$ [V] and $V_{\text{DS}}=0.1$ [V], and the $I_{\text{ON}}/I_{\text{OFF}}$ ratio= 5.35×10^6 .

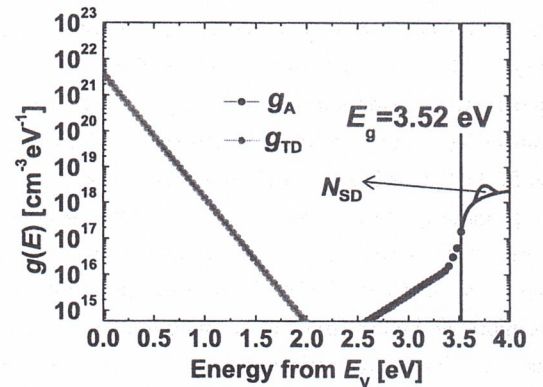


Fig. 3 The extracted subgap DOS of a-IGZO TFT.

The extracted subgap DOS of a-IGZO TFTs was shown in Fig. 3. First of all, the $g_A(E)$ was extracted by using the optical C - V spectroscopy [6]. It was found that the $g_A(E)$ was fitted very well with the superposition of two exponential formulae as follows:

$$g_A(E) = N_{DA} \times \exp\left(\frac{E - E_C}{kT_{DA}}\right) + N_{TA} \times \exp\left(\frac{E - E_C}{kT_{TA}}\right) \quad (1)$$

where the extracted parameters were $N_{TA} = 2.54 \times 10^{17} [\text{cm}^{-3}\text{eV}^{-1}]$, $kT_{TA} = 0.01 [\text{eV}]$, $N_{DA} = 2.84 \times 10^{16} [\text{cm}^{-3}\text{eV}^{-1}]$, and $kT_{DA} = 0.23 [\text{eV}]$. The value of N_{TA} was relatively lower than that of a-Si TFTs ($N_{TA} = 10^{21} [\text{cm}^{-3}\text{eV}^{-1}]$ for a-Si [19]). Subsequently, the extraction of $g_{TD}(E)$ was done by the I_{G-R} spectroscopy [7]. When the $g_{TD}(E)$ was assumed to be one exponential function, it could be modeled as

$$g_{TD}(E) = N_{TD} \times \exp\left(\frac{E_V - E}{kT_{TD}}\right) \quad (2)$$

where the extracted parameters were $N_{TD} = 4.0 \times 10^{21} [\text{cm}^{-3}\text{eV}^{-1}]$ and $kT_{TD} = 0.125 [\text{eV}]$. Finally, a technique of combining the optical $C-V$ spectroscopy [6] and the I_{G-R} spectroscopy [7] with the improved DeAOTS [8], [9] was used for the extraction of N_{SD} . The extracted value was $N_{SD} = 1.77 \times 10^{16} [\text{cm}^{-3}]$.

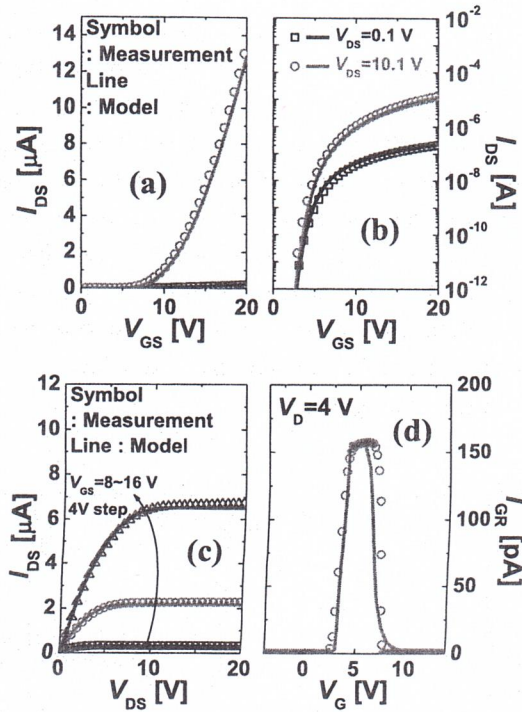


Fig. 4 The measured I - V characteristics of a-IGZO TFTs (symbols) in comparison with the calculated ones by DeAOTS (lines). (a) The transfer curve in a linear scale, (b) the transfer curve in a log scale, (c) the output curve, and (d) the I_{G-R} curve.

Based on the extracted DOS parameters, both the $I_{DS}(V_{GS}, V_{DS})$ and the $I_{G-R}(V_G, V_D)$ characteristics were

calculated by using DeAOTS [8], [9]. As shown in Fig. 4, the calculated $I_{DS}(V_{GS}, V_{DS})$ and $I_{G-R}(V_G, V_D)$ curves agreed very well with the measured ones. Our results verify that all of the extracted parameters, the implemented $I_{DS}(V_{GS}, V_{DS})$ and $I_{G-R}(V_G, V_D)$ models, and the parameter-extracting methodology can successfully reproduce the measured electrical characteristics even for the subthreshold region as well as for $V_{GS} > V_T$. Furthermore, the proposed method can be a main step triggering the process optimization, device characterization, and modeling for a systematic design and implementation of oxide TFT circuits and applications because DOS is the process-controlled parameter.

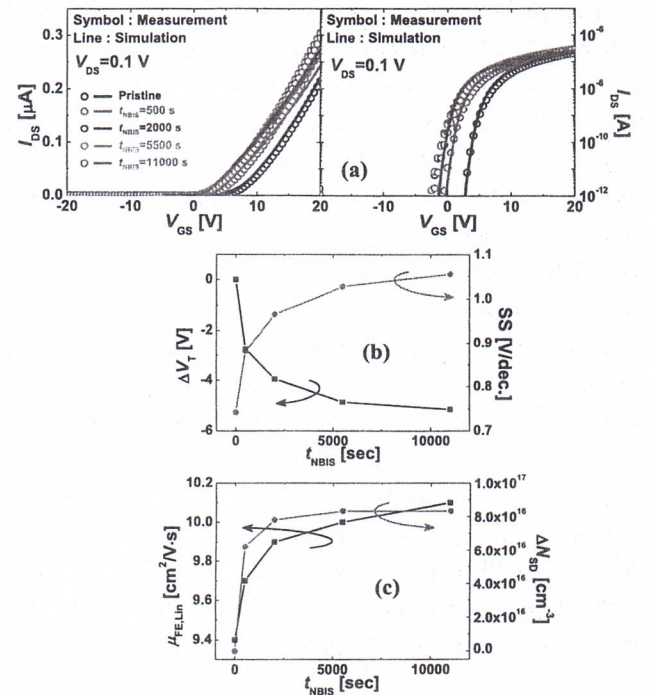


Fig. 5 (a) The measured t_{NBIS} -evolution of transfer characteristics (symbols) in comparison with the model (lines). (b) The measured t_{NBIS} -evolutions of ΔV_T and SS . (c) The t_{NBIS} -evolutions of the measured μ_{FE} (in linear operation regime) and the model-based ΔN_{SD} .

3. DOS-Based Instability Models

In order to establish the DOS-based instability model, the IGZO TFT I - V characteristics were measured with the increase of t_{NBIS} . The conditions of NBIS were $V_{GS} = -20 \text{ V}$, $V_{DS} = 10 \text{ V}$, and t_{NBIS} from 0 to 11000 sec, which are consistent with AMLCD switching TFTs during the most of each frame time. The illumination was applied by a commercial LED backlight unit with a brightness of 300

[cd/m²]. During the NBIS test, all the electrical properties were measured with halting the NBIS, by using Agilent 4156 semiconductor parameter analyzer under dark and ambient backlight illumination condition at room temperature. As shown in symbols of Fig. 5(a), the NBIS-induced ΔV_T without a significant change of SS is observed to be $-2.75/-3.95/-4.86/-5.15$ [V] after NBIS for 500/2000/5500/11000 sec, respectively. The measured t_{NBIS} -evolutions of ΔV_T , SS, and μ_{FE} (in linear operation regime) were also summarized in Fig. 5(b) and (c).

Regarding this widely observed NBIS-induced negative ΔV_T of IGZO TFTs, the valence band tail (VBT) hole (h^+)-intermediated formation of the O Frenkel-pair defect, in other words, peroxide state (O_2^{2-}), was very recently suggested as a microscopic origin for the NBS/NBIS-induced instability [10]. Because the VBT states of IGZO films have a $pp\sigma^*$ anti-bonding character, the anti-bonding state is empty by excited holes in $g_{\text{TD}}(E)$ under NBIS, and the photo-illumination makes the two oxygen atoms prefer to bond as O_2^{2-} . While the mechanism from the pristine to peroxide state via the transition state (see the inset of Fig. 6) can be denoted by $\text{O}^{2-} + \text{O}^{2-} + 2h^+ \rightarrow \text{O}_2^{2-}$ as well as $\text{O}^{2-} + \text{O}^{2-} + 2e^- + 2h^+ \rightarrow \text{O}_2^{2-} + 2e^-$ (electron doping), the recovery mechanism can be described as $\text{O}_2^{2-} + 2e^- \rightarrow \text{O}^{2-} + \text{O}^{2-}$ (from the peroxide to pristine state). Here, this peroxide state is the meta-stable state and is recovered very slowly after eliminating NBIS, which is due to the relatively higher energy barrier (β in [10]). By the process of $\text{O}^{2-} + \text{O}^{2-} + 2e^- + 2h^+ \rightarrow \text{O}_2^{2-} + 2e^-$ (electron doping), the formation of O_2^{2-} acts as the increase of (2+) donors and is followed by the parallel shift of IGZO TFT $I_{\text{DS}}-V_{\text{GS}}$ curves into the negative V_{GS} direction [as seen in Fig. 5(a)].

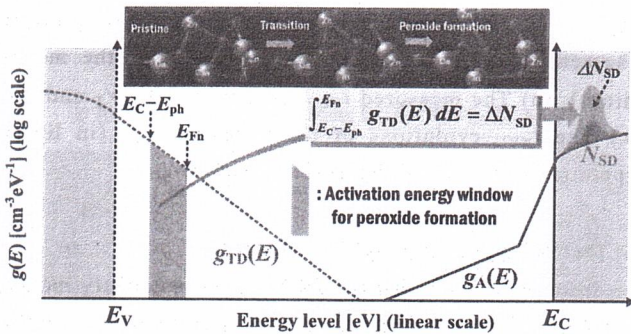


Fig. 6 The schematic diagram illustrating the increase of N_{SD} (ΔN_{SD}) resulting from the VBT hole-intermediated peroxide formation, which can be quantified by the activation energy window for peroxide formation in $g_{\text{TD}}(E)$ under NBIS. The inset shows the atomic structures

from the pristine to peroxide configurations via the transition calculated in a-IGZO. The equation in the yellow box suggests that the area of the activation energy window for peroxide formation is exactly the same as ΔN_{SD} .

Thus, we incorporated this microscopic view into the device simulation platform for instability-aware design. For a more quantitative model, we defined the area of

$\int_{E_C - E_{\text{ph}}}^{E_{\text{Fn}}} g_{\text{TD}}(E) dE$ as the “activation energy window for

peroxide formation” (Fig. 6), where E_{Fn} = the quasi-Fermi energy level and E_{ph} = the photon energy. According to the peroxide model, the IGZO TFT with a smaller activation energy window would be definitely more stable under NBIS and the increase of N_{SD} (ΔN_{SD}) resulting from the O_2^{2-} formation can be calculated by using the relation of

$$\Delta N_{\text{SD}} = \int_{E_C - E_{\text{ph}}}^{E_{\text{Fn}}} g_{\text{TD}}(E) dE \quad (\text{Fig. 6}) \quad \text{with all}$$

parameters-incorporating DeAOTS. Based on our model, we tried to trace the t_{NBIS} -evolution of $I-V$ characteristics by updating the stress time-varying DOS in the next stress time spot from one in the former time spot and repeating these update procedures time step by step (Fig. 7). With using $E_{\text{ph,eff}} = 2.49$ eV as empirical parameter, surprisingly, we found that the measured t_{NBIS} -evolution of $I_{\text{DS}}(V_{\text{GS}}, V_{\text{DS}})$ could be reproduced as seen in the lines of Fig. 5(a). The t_{NBIS} -evolution of the calculated N_{SD} was also shown in Fig. 5(c). With the increase of t_{NBIS} , the moving up of E_{Fn} (at NBIS condition) toward E_C becomes retarded, which supports the quantitative simulation of the saturated ΔV_T during NBIS.

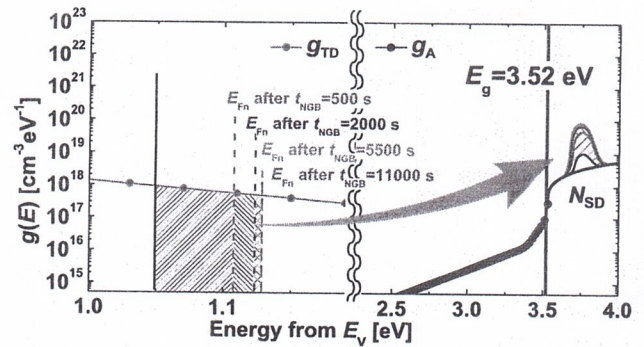


Fig. 7 The t_{NBIS} -evolution of the E_{Fn} with $E_{\text{ph,eff}} = 2.49$ eV and the activation energy window for peroxide formation.

This suggests that the t_{NBIS} -induced ΔV_T can be described with the exact function of not only the material

and process-controlled parameters, e.g., DOS, the device geometric parameters, e.g., T_{IGZO} and T_{OX} , but also the stress condition, e.g., gate bias, E_{Fn} at bias stress, and E_{ph} . For instance, the oxide TFT-A which has a relatively smaller area of the activation energy window shows the smaller stress-induced ΔV_{T} under specific stress conditions than that shown in the other oxide TFT-B which has relatively larger area of the activation energy window compared with that of TFT-A. The t_{NBIS} -evolution of either all electrical characteristics or the subgap DOS of a-IGZO TFTs can be also calculated precisely by using our methodology and parameter set.

4. Conclusions

We showed that the t_{NBIS} -evolution of the electrical characteristics of a-IGZO TFTs could be calculated and projected through our model based on the activation energy window of the VBT hole-intermediated peroxide formation. Therefore, our results have the potential of playing a powerful role in the instability-aware design of oxide TFTs without a long-term stress test, and also are consistent with the recently formulated O Frenkel-pair defect theory.

Acknowledgements

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